

# Preparation of Ti–Si mixed oxides by sol–gel one step hydrolysis

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## Abstract

Ti–Si mixed oxides of various  $\text{TiO}_2$  contents were prepared via sol–gel one step hydrolysis with economical  $\text{TiCl}_4$  and tetraethoxy silane (TEOS) as the precursors and characterized by means of  $\text{N}_2$ -physisorption, FT-IR and XRD. The effects of  $\text{TiO}_2$  content on the gelation time were studied. The influences of calcination temperature on the mixed oxide of 20 wt.%  $\text{TiO}_2$  (20TiSi) were also studied. The results indicate that titanium is in four-fold coordination with oxygen in the  $\text{SiO}_4^{4-}$  structure and Ti–O–Si band is observed by FT-IR. The mixed oxides with large  $\text{TiO}_2$  content exist on the pattern of amorphous material but not anatase. The composites have high specific surface areas, up to  $681.5 \text{ m}^2/\text{g}$ , and a narrow pore size distribution. The ratio of the micropore declines with the increase of  $\text{TiO}_2$  content. 20TiSi has good thermal stability and still possesses  $11 \text{ m}^2/\text{g}$  after calcination at  $1100^\circ\text{C}$ . Within the range of tested calcination temperature, Ti–O–Si band does not break.  $\text{TiO}_2$  is well dispersed as the form of amorphous in 20TiSi calcined at  $1100^\circ\text{C}$ .

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**Keywords:** Ti–Si mixed oxides; Sol–gel one step hydrolysis; Gelation time

## 1. Introduction

Titania–silica mixed oxide materials have received much attention over recent years because of their physical properties and lend them to a number of technological uses. These materials have been used as ultra-low-thermal-expansion glasses [1], catalysts, catalyst support [2,3] and thin films with tailored refractive indices [4]. The properties of titania–silica binaries are strongly dependent on their chemical composition, homogeneity, and texture, which in turn depend on synthesis conditions. Homogeneity at atomic level is especially important. And there are many methods to prepare Ti–Si mixed oxides, such as conventional solid-state mixing, coprecipitation and sol–gel process. Among these methods, sol–gel process is an effective route. Sol–gel synthesis based on the hydrolysis and subsequent condensation of metal alkoxide precursors provides a low temperature route to materials with a high level of atomic mixing and a high degree of porosity. When the sol–gel process is used to prepare titania–silica mixed materials, the different rates of hydrolysis of the silicon and titanium alkoxides can cause

phase separation, producing Ti- and Si-rich regions. In order to solve this problem, the researchers [5,6] took two kinds of routes. Firstly, the Si alkoxides partially hydrolyzed prior mixed with the Ti alkoxides. Secondly, Ti alkoxides reacted with acetylacetone in order to produce less reactive precursors which in turn react with prehydrolysed solution of Si alkoxides. The above measures complicate the procedure and titanium alkoxides make the cost increase. In this work we used a simple sol–gel method to prepare Ti–Si mixed oxides avoiding the prehydrolysis of silicon alkoxide and organic modification of titanium alkoxide.

Moreover, in a high temperature reactions, it is very important that the catalyst has good thermal stability. The study on the stability of catalysts becomes more and more important. For example, Yu et al. [7] studied the phase transformation under different calcination temperature, and Yan et al. [8] reviewed the effect of heat temperature on the Ti–Si composites.

In this paper, we used the cheap inorganic titanate— $\text{TiCl}_4$  as the starting material and prepared the Ti–Si mixed oxides by sol–gel one step hydrolysis process in the existence of acetic acid. The properties of the mixed oxides were studied by means of  $\text{N}_2$ -physisorption, XRD, FT-IR, etc. And the effects of calcination temperature on the Ti–Si mixed oxide of 20 wt.%  $\text{TiO}_2$  were also investigated.

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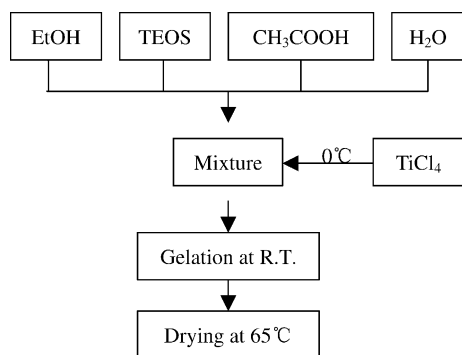


Fig. 1. The procedure for preparing Ti-Si mixed oxides.

## 2. Experimental

Ti-Si mixed oxides with different  $\text{TiO}_2$  contents were prepared with the following precursors: tetraethoxysilane (TEOS),  $\text{TiCl}_4$ , acetic acid ( $\text{CH}_3\text{COOH}$ ) and distilled water. And ethanol was used as a mutual solvent. The procedure was shown in Fig. 1. The amounts of TEOS and  $\text{TiCl}_4$  were varied to obtain mixed oxides with  $\text{TiO}_2$  content from 0 to 100 wt.%, and the corresponding molar ratios of  $\text{TEOS}:\text{TiCl}_4:\text{EtOH}:\text{H}_2\text{O}:\text{CH}_3\text{COOH}$  were  $x:y:7(x+y):10(x+y):2(x+y)$ . Firstly, EtOH, TEOS,  $\text{H}_2\text{O}$  and  $\text{CH}_3\text{COOH}$  were mixed by stirring, then  $\text{TiCl}_4$  was added to at  $0^\circ\text{C}$  under vigorous stirring. The resulting sol was stirred for a few minutes before gelling at room temperature and drying at  $65^\circ\text{C}$ . All samples were heated to  $400^\circ\text{C}$  in atmosphere and marked with  $x\text{TiSi}$ , in which  $x$  displayed the normal  $\text{TiO}_2$  content in weight percentage.

Moreover, the effects of varied calcination temperature were studied with 20TiSi. The specimen names were indicated as follows: 20TiSi-300, 20TiSi-400, 20TiSi-500, 20TiSi-700, 20TiSi-900, 20TiSi-1100. For comparison, 20TiSi-900 was prepared, in which  $\text{SiO}_2$  and  $\text{TiO}_2$  gained by sol-gel method were mixed by the weight ratio of 1:4 and were calcined at  $900^\circ\text{C}$  in atmosphere.

## 3. Characterization of the Ti-Si mixed oxides

### 3.1. Determination of textural properties

The specific surface area of the mixed oxides with different  $\text{TiO}_2$  weight content and 20TiSi calcined at different temperature were determined by nitrogen physisorption at 77 K using CE SORPTOMIC 1990 SERIES. The total pore volume (and the average diameter of pore, etc.) were obtained by the Barrett-Joyner-Halenda (BJH) method from adsorption/desorption curves of  $\text{N}_2$  at 77 K.

### 3.2. X-ray powder diffraction study

X-ray powder diffraction (XRD) patterns were measured on a Rigaku D/max 2500 X-ray diffractometer. The diffrac-

tograms were recorded with  $\text{Cu K}\alpha$  radiation over a  $2\theta$  range of  $10^\circ$ – $80^\circ$ .

### 3.3. FT-IR study

FT-IR measurements were performed on a 1730 Infrared Fourier Transform Spectrometer (Perkin-Elmer) using the potassium bromide as the background.

## 4. Results and discussion

### 4.1. Experiment of gelation time

Table 1 shows the gelation time of the materials with different  $\text{TiO}_2$  weight content. It is observed that the gelation time first decreases from 165 h (pure  $\text{SiO}_2$ ) to 8 h (20TiSi), then prolongs from 8 h (20TiSi) to 60 h (80TiSi) with the increasing of  $\text{TiO}_2$  content. Two factors determine the hydrolysis and condensation rates. First factor is the acetic acid, which promote the hydrolysis of TEOS; the second is the titanate formed by the chelate of acetic acid and titanium, which accelerate the condensation of TEOS [9]. With respect to acetic acid, it plays two roles in the system. One side, it acts as the catalyst to promote the hydrolysis of TEOS; on the other side, it chelate with titanium to form titanate, which accelerate the condensation of TEOS and retard the hydrolysis and condensation of  $\text{TiCl}_4$ . The acetic acid make these two hydrolysis and condensation processes compatible. When a lower  $\text{TiO}_2$  is added to the system of TEOS, such as 10TiSi, the gelation time dramatically decreased, which consent to that titanate can speed up condensation of TEOS [9]. While when  $\text{TiO}_2$  content are increased further, more acetic acid is chelate with titanium than to catalyze the hydrolysis, which results in some extent the prolong of the gelation time. It is noticeable that  $\text{TiO}_2$  prepared from  $\text{TiCl}_4$  does not gelate. Generally,  $\text{TiO}_2$  gelate when prepared from titanium alkoxide by sol-gel method. So far, as we know, no one has studied the gelation time of  $\text{TiO}_2$  in the system of  $\text{TiCl}_4$ . We found that  $\text{TiO}_2$  could not gelate at  $30^\circ\text{C}$  but existed in the form of transparent sol. A lot of changes will happen along with the temperature rise and some precipitate will appear at  $75^\circ\text{C}$ . But the same phenomenon did not appear when  $\text{TiCl}_4$  was

Table 1  
The effect of  $\text{TiO}_2$  content on the gelation time of Ti-Si mixed oxides

| Samples        | Gelation (time/h) |
|----------------|-------------------|
| $\text{SiO}_2$ | 165               |
| 10TiSi         | 18.5              |
| 20TiSi         | 8                 |
| 40TiSi         | 31                |
| 60TiSi         | 39.5              |
| 80TiSi         | 60                |
| $\text{TiO}_2$ | $\infty$          |

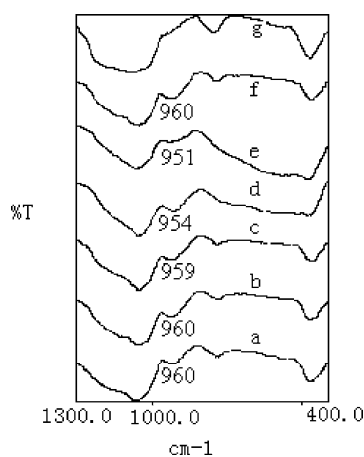


Fig. 2. The FT-IR of the mixed oxides. (a) SiO<sub>2</sub>, (b) 20TiSi, (c) 40TiSi, (d) 60TiSi, (e) 80TiSi, (f) 20TiSi-1000, and (g) SiO<sub>2</sub>-600.

replaced by Ti-alkoxide. From the above information, it can be concluded that the absence of TiO<sub>2</sub> gel is only because of the property of TiCl<sub>4</sub>. In addition, the amount of acetic acid have a big effect on the gelation time. The gelation time of 20TiSi and 60TiSi decreased when the amount of acetic acid increased and 60TiSi decreased more rapidly than 20TiSi did. When the molar ratio of CH<sub>3</sub>COOH:(TEOS + TiCl<sub>4</sub>) reached 7, 20TiSi and 60TiSi became gel almost at the same time. The further mechanism is still under investigation.

#### 4.2. Characterization bonds

Fig. 2 shows the FT-IR spectra of all the Ti–Si mixed oxides. The peaks in the spectra have been assigned according to the literature [6]. The spectra reveal three major absorption bands at about 1070, 950 and 460 cm<sup>−1</sup> (Fig. 2b–e), which represent the characteristic bonds of Si–O–Si asymmetric stretching (near 1070 cm<sup>−1</sup>), Ti–O–Si asymmetric stretching (around 950 cm<sup>−1</sup>) and Ti–O stretching (near 460 cm<sup>−1</sup>), respectively. The absorption band observed at about 950 cm<sup>−1</sup> is associated with titanium in four-fold coordination with oxygen in the SiO<sub>4</sub><sup>4−</sup> structure. SiO<sub>2</sub> also gives a 950 cm<sup>−1</sup> peak in the FT-IR spectrum, which is attributed to Si–OH

band. However, this band disappeared after heat treatment at 600 °C (Fig. 2g). In addition, the 950 cm<sup>−1</sup> absorption band is still observed on the 20TiSi, which has been thermally treated at 1000 °C (Fig. 2f). Accordingly, the 950 cm<sup>−1</sup> peak of Ti–O–Si mixed oxides should not be attributed to Si–OH band, but associated with Si–O–Ti band. Particularly, the characteristic bond was still existed when the content of TiO<sub>2</sub> is up to 80 wt.%. From the results, we could reach an important conclusion that titanium is in four-fold coordination with oxygen in the SiO<sub>4</sub><sup>4−</sup> structure, and each component in the materials is mixed on an atomic scale.

#### 4.3. Nitrogen physisorption results

##### 4.3.1. Specific surface areas and pore volumes

The results of the sorption measurements are listed in Table 2. Note that these mixed materials have high specific surface area, can reach to 549.3 m<sup>2</sup>/g (10TiSi), which is higher than the Ref. [5].

The specific surface area of the samples decreased from 549.3 m<sup>2</sup>/g (10TiSi) to 194.0 m<sup>2</sup>/g (80TiSi) when TiO<sub>2</sub> weight content increased and it is not dramatically decreased when TiO<sub>2</sub> content is not more than 20%, and the reason is that a little content of TiO<sub>2</sub> cannot result in the change of SiO<sub>2</sub> structure, which is consistent well with the XRD analysis. The decrease in surface area with TiO<sub>2</sub> content increasing can be partly explained by the higher atomic weight of Ti, but the loss of the most surface area must be attributed to an increased contribution of the pure titania character in the high-titania materials. According to the results shown in Table 2, pore specific volume (*V<sub>p</sub>*) of the specimen first decrease and then increase with along the rise of the TiO<sub>2</sub> content, and reach to the maxima at 20TiSi. It can be explained that the hydrolysis rate of TEOS and TiCl<sub>4</sub> is well compactable just as shown in the experiment of gelation time.

##### 4.3.2. Pore structure

Fig. 3 shows the nitrogen adsorption isothermal curves at 77.15 K for the samples prepared by sol–gel one step hydrolysis method. The isothermal adsorption of TiO<sub>2</sub> is type

Table 2  
Physisorption results of mixed oxides

| Samples             | Surface area BET (m <sup>2</sup> /g) | <i>V<sub>p</sub></i> (cm <sup>3</sup> /g) | Micropore (%) | Pore-size <sup>a</sup> diameter maxima (nm) | Pore-size diameter range (nm) |
|---------------------|--------------------------------------|---|---------------|---|-------------------------------|
| SiO <sub>2</sub>    | 582.04                               | 3.45                                      | 65.95         | 0.66  | 0.55–4.5                      |
| 10TiSi              | 549.3                                | 0.29                                      | 66.32         | 0.85  | 0.62–4.5                      |
| 20TiSi              | 511.5                                | 0.39                                      | 57.77         | 0.94  | 0.58–5.5                      |
| 40TiSi              | 219.6                                | 0.19                                      | 32.38         | 1.2   | 0.76–18.9                     |
| 60TiSi              | 292.0                                | 0.21                                      | 37.77         | 1.1   | 0.65–18.9                     |
| 60TiSi <sup>b</sup> | 63.8                                 | 0.10                                      | 23.62         | 1.2   | 0.87–40.6                     |
| 80TiSi              | 194.0                                | 0.18                                      | 19.86         | 1.3   | 0.82–16.0                     |
| TiO <sub>2</sub>    | 59.5                                 | 0.27                                      | 13.91         |   |                               |

<sup>a</sup> Maximum of pore size distribution.

<sup>b</sup> 60TiSi preparing with lower acetic acid-amount.

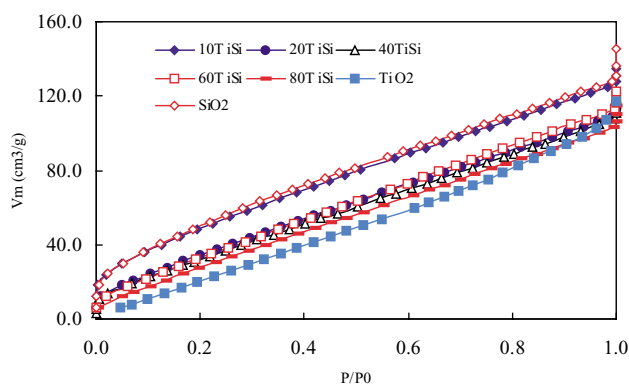


Fig. 3. N<sub>2</sub> physisorption isotherm of the mixed oxides.

III of the IUPAC isothermal classifications, indicating that TiO<sub>2</sub> contained both mesopores and macropores. The adsorption curves of SiO<sub>2</sub> and other mixed oxides are between types I and II isotherms, indicating that these specimens are composed of micropores, mesopores and macropores. Depending on the titania content, the tested materials reveal pore-size maxima between 0.85 and 1.3 nm (Table 2). No clear correlation is found between titania content and pore-size, but the pore distribution tends to be broad. The change of micropore ratios are in good agreement with those obtained from analysis of isothermal adsorption curves.

#### 4.4. Phase analysis

The XRD analysis results are given in Fig. 4. When the titania content is less than 80 wt.%, there is a weak and broad peak around  $2\theta = 25^\circ$ , which represented that these specimens prepared by sol-gel one step hydrolysis method and obtained after being thermally treated at 400 °C were all noncrystalline. These results suggest that the low titania materials are mixed on an atomic scale, which prevented the materials from phase separation and crystallization even at high content of TiO<sub>2</sub>. When the titania content reaches

80 wt.% the broad diffraction lines are located in  $25.5^\circ$ ,  $38.1^\circ$ ,  $48.2^\circ$ ,  $53.9^\circ$ ,  $55.4^\circ$  and  $62.8^\circ$  ( $2\theta$ ). They are characteristic lines of anatase TiO<sub>2</sub>. However, it is broader than pure TiO<sub>2</sub>, which proves that the size of anatase crystal in 80TiSi is very small. The absence of large sharp diffraction peak in TiO<sub>2</sub> rich samples indicated that the titanium in the samples did not exist as crystals of sufficient size to be detected by XRD and the relatively small proportion of silica is highly dispersed through the titania network homogeneously. The SiO<sub>2</sub> does restrain the crystallization of TiO<sub>2</sub> in the mixed oxides while this strong retarding has been often ascribed to a good chemical homogeneity of the starting gels, i.e., to a high degree of Si–O–Ti bonding, titania oxide species are dispersed in amorphous structures of SiO<sub>2</sub> matrices.

#### 4.5. Effect of calcination temperature

Fig. 5 shows the nitrogen adsorption curves at 77.15 K for 20TiSi calcined at from 300 to 1100 °C. The isothermal adsorption of 20TiSi calcined below 500 °C are between types I and II of IUPAC classification, indicating that these materials are composed of micropore, mesopore and macropore. When 20TiSi is calcined at 700, 900 °C, the adsorption are type III, showing that 20TiSi-700 and 20TiSi-900 have many macropore. The isotherm of 20TiSi-1100 is a beeline, which indicates that it is a kind of nonpore material.

It can be learned from physisorption results (Table 3) that with the elevation of temperature the specific surface area first increase, reach to the maxima at 400 °C, and then decrease.  $V_p$  and micropore volume have the same trend. This trend can be explained as follows: one hand, at lower temperature the samples will contain unreacted alkoxide groups, solvent molecules and hydroxyl group, and these groups will be decomposed when the temperature rising, which result in the increase of specific surface,  $V_p$  and micropore volume; on the other hand, with the elevation of the temperature, calcinations cause shrinkage of the structural pore. Further-

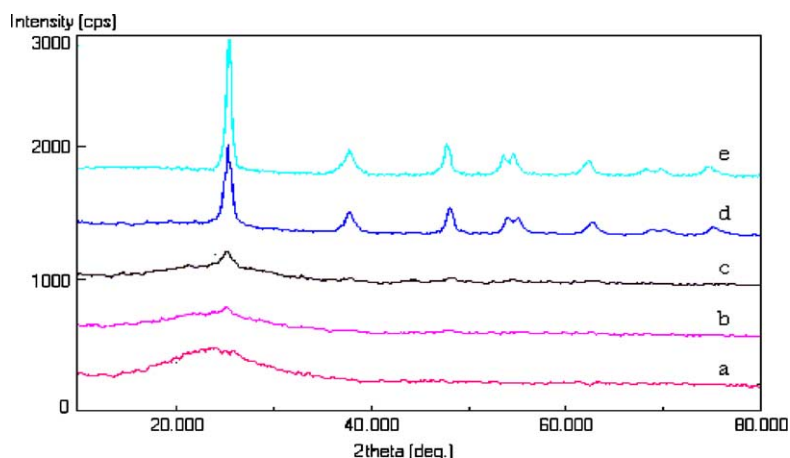
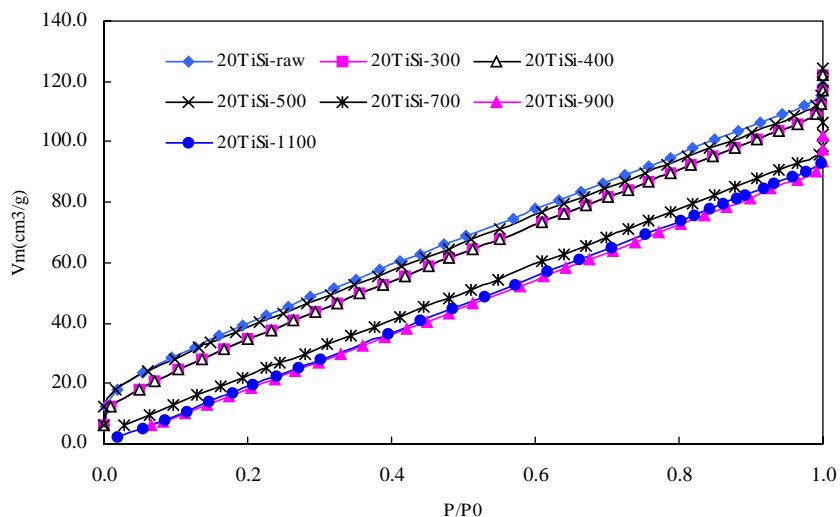


Fig. 4. XRD patterns of the mixed oxides. (a) 20TiSi, (b) 40TiSi, (c) 60TiSi, (d) 80TiSi, and (e) TiO<sub>2</sub>.

Fig. 5. N<sub>2</sub> physisorption isotherm of the mixed oxides.Table 3  
Physisorption results

| Samples     | Surface area<br>BET (m <sup>2</sup> /g) | V <sub>p</sub> BET<br>(cm <sup>3</sup> /g) | Micropore volume<br>(Dobinin) (cm <sup>3</sup> /g) |
|-------------|---|--|--|
| 20TiSi-300  | 422.9                                   | 0.26                                       | 0.19   |
| 20TiSi-400  | 511.5                                   | 0.39                                       | 0.24   |
| 20TiSi-500  | 293.2                                   | 0.21                                       | 0.15   |
| 20TiSi-700  | 90.5                                    | 0.10                                       | 0.04   |
| 20TiSi-900  | 66.4                                    | 0.11                                       | 0  |
| 20TiSi-1100 | 11                                      | 0.22                                       | 0  |

more, compared with the Ref. [10], 20TiSi has high specific surface area after calcination at 1100 °C.

The effect of the calcination temperature on the structure of TiO<sub>2</sub>–SiO<sub>2</sub>, revealed by XRD, is illustrated in Fig. 6. When 20TiSi is calcined at from 300 to 1100 °C, it is shown that there are only weak and broad diffraction around  $2\theta$

= 25°, which is characteristic of amorphous silica. This suggests that TiO<sub>2</sub> in the composite is highly dispersed. To compare with TiSi material the TiO<sub>2</sub>–SiO<sub>2</sub> mixed material were also calcined at 900 °C. Shown as (f), SiO<sub>2</sub> and TiO<sub>2</sub> in 20TiSi-900 appear in the pattern of amorphous and anatase respectively. Yu et al. [7] prepared Ti–Si composite particles which existed in the form of anatase at 473 °C and transformed into rutile at 1100 °C. Yan et al. [8] gained TiO<sub>2</sub>–SiO<sub>2</sub> mixed materials in which anatase appeared at 400 °C. Accordingly, it is suggested that TiO<sub>2</sub> and SiO<sub>2</sub> are mixed uniformly on an atomic scale in the sol–gel one step hydrolysis preparation. This atomic scale mixing retards the phase separation and crystallization even at elevated temperatures. Although the calcination may break the Ti–O–Si band [6], the mobility of the constitutional elements in the structure are not sufficiently high, so that the materials obtained at 1100 °C are still not crystal.

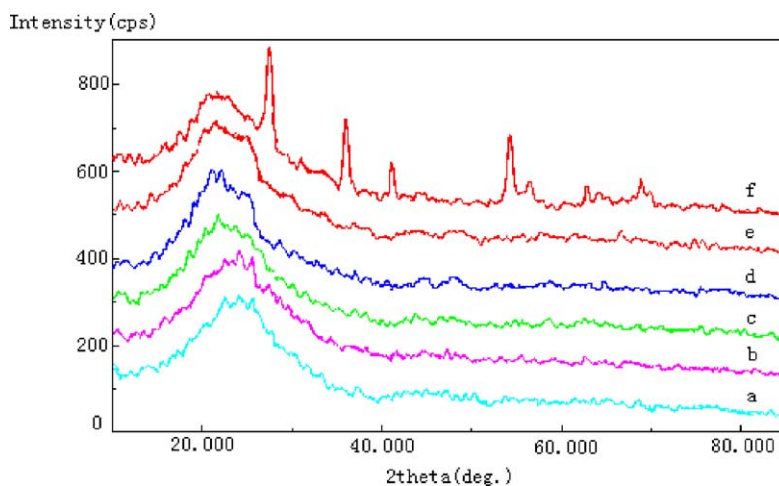


Fig. 6. XRD patterns of 20TiSi calcined at different temperature. (a) 20TiSi-300, (b) 20TiSi-500, (c) 20TiSi-700, (d) 20TiSi-900, (e) 20TiSi-1100, and (f) 20TiSi-900.



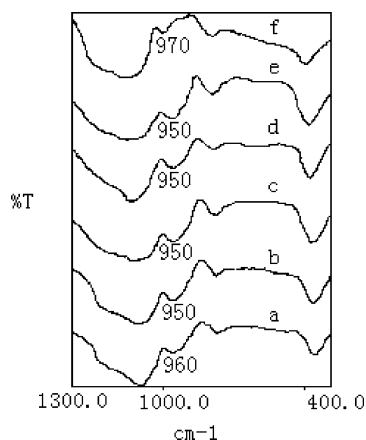


Fig. 7. FT-IR spectra of 20TiSi calcined at different temperature. (a) 20TiSi-300, (b) 20TiSi-500, (c) 20TiSi-700, (d) 20TiSi-900, (e) 20TiSi-1100, and (f) 20TiSi-900.

FT-IR of 20TiSi calcined at different temperature is shown in Fig. 7. The results indicate that the characteristic bond of  $950\text{ cm}^{-1}$  cannot vary in intensity and position when the calcination temperature increases, and calcination at  $1100^\circ\text{C}$  does not cause the break of Ti–O–Si. However, there is not the adsorption band of  $950\text{ cm}^{-1}$  but  $970\text{ cm}^{-1}$  in 20TiSi-900. Accordingly, it can be concluded that 20TiSi prepared by sol–gel one step hydrolysis method has good thermal stability, and calcination at  $1100^\circ\text{C}$  does not result in the break of Ti–O–Si.

## 5. Conclusion

Ti–Si composite materials are mixed on an atomic scale, and Ti–O–Si band appears in the all samples with different  $\text{TiO}_2$  content

1. The specimens have high specific surface area, up to  $549.3\text{ m}^2/\text{g}$ , and the adsorption isotherm of the mixed oxides are between types I and II.
2. The mixed oxides mainly remain amorphous pattern. Anatase appears when the  $\text{TiO}_2$  content is more than 80 wt.%, comparing with the intensity of pure  $\text{TiO}_2$ , it is very tiny.
3. 20TiSi has good thermal stability. It still possesses  $11.8\text{ m}^2/\text{g}$  after calcination at  $1100^\circ\text{C}$ . Calcination on the large scale of temperature does not cause the break of Ti–O–Si band.  $\text{TiO}_2$  is well dispersed in 20TiSi calcined at  $1100^\circ\text{C}$ , and still existed in the form of amorphous.

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